Deposition of Europium Oxide on Si and its optical properties depending on thermal annealing conditions

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Abstract We investigate the influence of the ambient gas during thermal annealing on the photoluminescence (PL) properties of europium compound thin films on Si substrates. The films were deposited by radio-frequency magnetron sputtering and subsequently annealed in N₂ or O₂ ambient gas by rapid thermal annealing (RTA). The results of X-ray diffraction indicate that the resulting europium compound annealed in N₂ ambient have several silicate phases such as EuSiO₃ and Eu₂SiO₄ compared to those annealed in O₂ ambient. The spectral results revealed that a broad luminescence associated with Eu²⁺ ions, with a maximum intensity at 600 nm and a FWHM of 110 nm, was observed from the thin film annealed at 1000 °C in N₂ ambient. However, a series of narrow PL spectra from Eu³⁺ ions were observed from the film annealed in O₂ ambient.

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1 Introduction

Rare-earth silicate compounds have shown excellent luminescent properties [1, 2], and due to the thermal stability in optical transitions, they have been applied to lasers and optical fibers. In particular, europium-silicate compounds are materials that show intense and stable luminescence in the visible region. Several works have been done so far regarding the structure and the electrical/optical/magnetic properties of europium-silicate compounds [3-6]. Pure europium silicate is a kind of material known as a magnetic semiconductor. However, the growth of single-phased europium silicate crystals from EuO and SiO₂ generally requires temperatures as high as 1800 °C for Eu₂SiO₄ [4] and 1400 °C for EuSiO₄ [5]. Due to difficulties in growing pure crystals, a few reports on the applications of europium silicates [6] have been made compared to other europiumdoped materials in the visible region. Phosphor doped with Eu³⁺ give rise to a strong red emission at approximately 610 nm, and they play an important role in emissive display technology such as plasma display panel (PDP) and fieldemission display, and in the lighting industry such as fluorescent lamps [7–9]. In addition, Eu²⁺ ions in the host lattice emit UV to red light under UV irradiation. Therefore, Eu^{2+} is an attractive phosphor, especially in the blue region, for display technology [10–11].

In this work, we report on the formation and the optical characteristics of europium-compound thin films deposited by using an RF-sputtering method. The samples were annealed in N_2 or O_2 ambient gases by rapid thermal

annealing (RTA) and were characterized by using X-ray diffraction (XRD) spectroscopy, Auger electron spectroscopy (AES), transmission electron microscopy (TEM) and X-ray photoelectron spectroscopy (XPS). The photoluminescence (PL) properties of the europium-compound thin films were investigated.

2 Experiments

Europium-oxide thin films were prepared on *p*-type Si (100) substrates at room temperature by using an RF sputtering method in Ar gas with a Eu₂O₃ (99.95%) target. During the sputtering process, the atmosphere gas pressure was kept at 4.4 mTorr, Ar flow rate was 50 sccm, and the RF power was 70 W. The deposited film thickness was about 0.17 μ m. The europium-oxide films were annealed in the temperature range of 1000 °C for 5 min in N₂ or O₂ ambient by using RTA.

The films were characterized by using XRD, AES, TEM and XPS. The XRD patterns were measured by using a Rigaku DMAX2500 X-ray diffractometer with Cu K_{α} (1.542 Å) radiation as the incident X-ray source to identify the formation of europium silicates based on the literature profiles of the Joint Committee on Powder Diffraction Standards (JCPDS). The sputtered films were depth profiled with AES (PHI-680 Auger Nanoprobe) in order to determine the compositional distribution as a function of depth in the film. The TEM studies were performed using JEOL JEM3010 electron microscope. The XPS measurements were carried out in a Perkin-Elmer Model PHI 5800 ESCA system with a monochromatic X-ray source of Al K α (1,486.6 eV). The spectrometer was operated at a constant pass energy of 187.85 eV. An energy step of 0.125 eV was used to acquire the XPS spectra. The PL emission spectra of the samples were obtained using a 325 nm He-Cd laser at room temperature.

3 Results

Figure 1 shows the XRD patterns of the europium compound thin films, one of which was annealed in N_2 ambient and the other was annealed in O_2 ambient at 1000 °C for 5 min by using RTA. The XRD spectrum of the sample annealed at 1000 °C in N_2 ambient shows that EuSiO₂ and Eu₂SiO₄ are formed However, the XRD spectrum of the sample annealed in O_2 ambient shows a diffraction peak only from the Eu₂O₃ in the film as shown in Fig. 1(b). It is evident from these data that oxygen gas might restrain the formation of europium-silicate.

AES depth profiles obtained of europium compound thin films before and after thermal annealing in N_2 ambient are



Fig. 1 XRD patterns of the samples annealed at 1000 °C for 5 min in (a) N_2 ambient and (b) O_2 ambient

shown in Fig. 2. Compositional distribution of Eu and O of as-deposited thin film is shown in Fig. 2(a). After thermal annealing at 1000 °C in nitrogen ambient, depth profiles of the thin film are changed as shown in Fig. 2(b). The variations of the atomic concentrations show that some Si atoms are diffused into the deposited thin film out of the substrate while Eu and O are diffused into the substrate, leading to a composite film of Eu, O and Si. This indicates that the thin film forms europium-silicates after thermal annealing, which coincides with the XRD results. However, Fig. 3(a) shows different compositional profiles of the thin film after thermal annealing at 1000 °C in oxygen ambient. The thin film is composed of Eu and O. We confirm from the Auger spectrum that europium-silicates are rarely formed in the film during the annealing in O₂ ambient. It is thought to be due to the SiO₂ film formed by intermixing of Si coming from the substrate and oxygen coming from the thin film, preventing the intermixing of Si and Eu atoms. It was also verified by transmission electron microscopy. Figure 3(b) shows a cross-sectional TEM image of the interface between the europium oxide layer and silicon substrate in the sample which was annealed in O₂ ambient. It is seen that the silicon oxide layer at the interface is formed.

The samples have been studied by the XPS to investigate the chemical binding states of the constituent atoms in the thin films. The peaks monitored in these spectra correspond



Fig. 2 Auger depth profiles for (a) the "as deposited" sample, and (b) the sample annealed at 1000 °C for 5 min in N_2 ambient

to Si 2p, O 1s, and Eu 3d core levels. The XPS spectra of the sample annealed in nitrogen ambient are shown in Fig. 4. In the Si 2p spectrum [Fig. 4(a)] the spectral peak located at a binding energy of 99.6 eV is attributed to the



Fig. 3 (a) Auger depth profile and (b) cross-sectional TEM image for the sample annealed at 1000 $^\circ$ C for 5 min in O₂ ambient

Si. In the O 1s spectrum [Fig. 4(b)] the spectral peak at the binding energy of 531.8 eV is considered to originate from silicate compounds. The main peak at 533.0 eV is due to the contribution of silicon oxide. In the Eu 3d (5/2) spectrum [Fig. 4(c)], both for Eu^{2+} and Eu^{3+} systems the 3d(5/2) line is split into two compounds (the main peak at a binding energy of 1,128.3 eV corresponds to the divalent europium ions; while the binding energy peak at about 1,137.0 eV originates from the trivalent europium ions together with the contribution of the shake-up of divalent europium ions). The shape and widths of the two components are different, which can be expected due to multiplet splitting for two different final-state 4f configurations. In a similar manner, the strong peak at the binding energy of 1,156.7 eV in the Eu 3d (3/2) spectrum corresponds to the divalent europium ions. No trivalent europium state was observed in the XPS spectra of the sample. The spectrum of the Eu 3d (5/2) core level is similar to the spectrum of typical



Fig. 4 XPS spectra of (a) Si 2p, (b) O 1s, and (c) Eu 3d for the samples annealed at 1000 $^\circ C$ for 5 min in N_2 ambient

divalent (Eu metal) europium compounds [12]. Since the shape and the binding energy position of the Eu 3d (5/2) peak agree with those of Eu metal, we can conclude that the Eu ion exists in the divalent state in this film.

The XPS spectra of Si 2p, O 1s and Eu 3d core levels of the sample annealed in oxygen ambient are shown in Fig. 5. However, Si 2p peak is not detected because of its low concentration in the film. In the XPS spectrum of O 1s [Fig. 5(b)] the spectral peak is the contribution from europium oxides at the binding energy of 528.6 eV. No europium silicate peak in Fig. 5(b) exists compared to that of the sample annealed in N₂ ambient [Fig. 4(b)]. In the spectrum of Eu 3d [5/2; Fig. 5(b)] the spectral peak at a binding energy of 1,124.0 eV corresponds to the divalent europium ions, while the binding energy peak at about 1,132.8 eV originates from the trivalent europium ions together with the contribution of the shake-up of divalent europium ions. Obviously, the spectral shapes of the sample



Fig. 5 XPS spectra of (a) Si 2p, (b) O 1s, and (c) Eu 3d for the samples annealed at 1000 $^{\circ}$ C for 5 min in O₂ ambient



Fig. 6 Photoluminescence emission spectra with an excitation wavelength of 325 nm (He-Cd laser) from the samples annealed at 1000 °C for 5 min in (a) N_2 ambient (b) O_2 ambient

annealed in O_2 ambient are different from those of the sample annealed in N_2 ambient. Much strong spectral peak of trivalent europium state was observed in the XPS spectra of the sample annealed in O_2 ambient than that of the sample annealed in N_2 ambient. The spectrum of the Eu 3d (5/2) core level is along with the spectrum of typical trivalent (Eu₂O₃) europium compounds [12]. It shows that the shape and the binding energy position of the Eu 3d (5/2) peak agree with those of Eu₂O₃.

Figure 6 shows the emission spectra from the europiumcompound thin films annealed at different temperature in nitrogen or oxygen ambient. Figure 6(a) shows the broadband PL spectra from the thin films, with the maximum intensity at 600 nm and a full width at maximum of 110 nm, annealed at 1000 °C in nitrogen ambient. This broad spectrum is thought to originate from the $4f^{6}5d-4f^{7}$ transition of divalent europium ions Eu^{2+} at different sites. The intense and broad luminescent properties of the europium-silicate thin films are expected to have wide applications in areas of passive optical components, like thin film phosphors, and active components, like electroluminescence devices. By contrast, in Fig. 6(b), a series of narrow PL spectra are observed from the film annealed in O_2 ambient, which is thought to originate from the $5D^J - 7F^J$ transition energy of Eu³⁺ ions in solid matrices.

4 Conclusion

Europium-compound thin films on silicon substrates were fabricated by radio-frequency magnetron sputtering and subsequent annealing. We investigated the influence of the ambient gas during thermal annealing on the PL properties and studied the films by XRD, AES, TEM and XPS. The results of X-ray diffraction indicated that the resulting europium compound annealed in N2 ambient have several silicate phases such as EuSiO₃ and Eu₂SiO₄ compared to those annealed in O₂ ambient. The spectral results revealed that a broad luminescence associated with Eu²⁺ ions, with a maximum intensity at 600 nm and a FWHM of 110 nm, was observed from the thin film annealed at 1000 °C in N₂ ambient. However, a series of narrow PL spectra from Eu³⁺ ions were observed from the film annealed in O2 ambient due to the SiO₂ diffusion barrier produced between Eu₂O₃ and Si substrates.

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